Nonradiative Energy Transfer and Photocurrent Enhancements in Hybrid Quantum Dot-MoS₂ Devices

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ABSTRACT

The energy transfer mechanism leading to highly efficient nonradiative energy transfer (NRET) from quantum dots (QDs) to monolayer MoS₂ devices has been investigated *via* a spectral dependence study. The spectral dependence of the NRET from three spectrally separated QD ensembles to monolayer MoS₂ devices reveals that the trends in NRET rates follow the trends in the spectral overlap between the QD emission and MoS₂ absorption spectra, thus verifying that the mechanism for the energy transfer is Förster-type NRET. Furthermore, the dependence of the photocurrent enhancement on the MoS₂ film properties is also explored.

Keywords: nonradiative energy transfer, quantum dots, 2D materials, photocurrent, photoresponsivity.

1. INTRODUCTION

In recent years there has been a significant focus on research into 2D materials due to their unique optical and electrical properties. It was the discovery and isolation of graphene that initially sparked interest in 2D materials [1]. Graphene, however, lacks a direct bandgap which is a desirable property when considering optoelectronic devices. Transition metal dichalcogenides (TMDs) such as MoS2, WS2, WSe2 and MoSe2 can also be isolated into monolayers and have the advantage of a transition to direct bandgap semiconductors at monolayer thicknesses [2]. The presence of a direct bandgap at monolayer thickness gives these TMDs the edge over graphene in terms of optoelectronic devices. However, when considering an optoelectronic device architecture that consists of monolayer materials, the low levels of optical absorption must be taken into account. The inclusion of a sensitizing layer is beneficial in terms of enhancing the optical absorption in the hybrid system and must facilitate the efficient transfer of energy from the sensitizer to the monolayer material. One particular mechanism of energy transfer from the sensitizing species to the active material is Förster-type nonradiative energy transfer (NRET). This has been proven to be a highly efficient process in donor-acceptor pairs of quantum dots (QDs) and monolayer TMDs [3]-[5]. It has also been revealed that, in contrast to graphene, the NRET rate between QDs and MoS2 decreases as the layer number of the material increases. This behaviour is a consequence of the strong increase in dielectric screening with the addition of each MoS₂ layer [4]. The incorporation of a sensitizing species on MoS₂ photodetector devices has been shown to substantially increase the photocurrent and photoresponsivity of the devices [5]-[7]. In this contribution it is confirmed that the energy transfer mechanism in QD-sensitised MoS₂ devices is Förster-type NRET. This has been probed via a spectral dependence study of NRET to monolayer MoS₂ device using three QD ensembles with emission wavelengths spanning different regions of the visible spectrum. The effect of the CVD-grown MoS₂ film properties on the photocurrent enhancements is also reported.

2. EXPERIMENTAL METHODS

MoS₂ of various film properties and thickness was grown on a Si/SiO₂ substrate using a large area coverage chemical vapor deposition (CVD) growth technique [8].

Alloyed core/shell CdSeS/ZnS QDs were purchased from Sigma-Aldrich in toluene (1 mg/mL). In order to form thin film layers the QDs were dispersed in 0.1% wt. PMMA in toluene. The QD/PMMA solutions were prepared by dispersing 12.5 μ L of stock QD solution in 500 μ L of 0.1% wt. PMMA in toluene. The dispersions were sonicated for ~20 s to ensure even distribution of the QDs in the PMMA matrix. The alloyed QDs with peak emission wavelengths of 450 nm, 530 nm, and 630 nm have an average diameter of (6 \pm 0.8) nm.

The MoS₂ devices were fabricated *via* the patterning of a PMMA e-beam resist using electron beam lithography. The contact pads and electrodes consisting of Ti/Au (5 nm/45 nm) were deposited using an e-beam evaporation technique. For the spectral dependence study, the device samples were grown on different Si/SiO₂ chips for each of the QD ensembles during a single growth process. For the MoS₂ film properties study, all devices were grown on the same Si/SiO₂ chip in a single growth process. The hybrid QD-MoS₂ devices were fabricated by spin coating an ultrathin layer of QDs in PMMA on top of the MoS₂ devices to obtain monolayer coverage of QDs.

Time-resolved PL (TRPL) measurements were performed using an excitation with 90 ps pulses at a wavelength of 405 nm and a repetition rate of 10 MHz. The sample was excited through a $40 \times$ objective with an NA of 0.65 and a spot size of ~430 nm. The PL was collected through the same objective.

The photocurrent measurements were performed using a variable power laser diode with an excitation wavelength of 405 nm. The active region in the device was excited through a $10 \times$ objective with an NA of 0.25 and a spot size of $\sim 3 \mu m$. A Keithley 2400 source meter was used to provide a bias and measure the current flowing through the devices.

3. RESULTS AND DISCUSSION

The extinction spectrum of monolayer MoS₂ and the extinction spectra of each of the QD ensembles are shown in Fig. 1(a), along with the normalized PL spectra of the QDs. There is good overlap between the 630 nm QDs emission spectrum and the 'B' exciton in the MoS₂ extinction spectrum. The emission spectrum of the 450 nm QDs also overlaps quite well with the high energy 'C' exciton in the MoS₂ extinction spectrum, while there is no overlap between the emission spectrum of the 530 nm QDs with any of the MoS₂ exciton peaks. A schematic representation of the hybrid devices is given in Fig. 1(b).

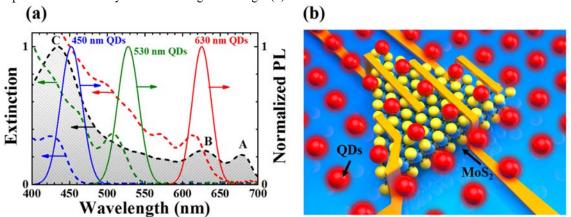


Figure 1: (a) Extinction spectra (left axis) of monolayer MoS₂ (black dash), 450 nm QDs (blue dash), 530 nm QDs (green dash), and 630 nm QDs (red dash), and normalized PL spectra (right axis) of 450 nm QDs (blue solid), 530 nm QDs (green solid), and 630 nm QDs (red solid). The QD extinction spectra in (a) have been scaled to give the same intensity at the first excitonic peak. (b) Schematic representation of the hybrid QD-MoS₂ devices.

TRPL measurements were performed to measure the lifetime of the QDs on the devices, τ_{QD-MoS_2} , and the lifetime of the QDs off the devices, τ_{QD} . The PL decays and lifetimes of the 450 nm QDs (blue), 530 nm (green), and 630 nm QDs (red) are shown in Fig. 2, panels (a-c), respectively. The black curves in Fig. 2(a-c) are the PL decays of the QDs on the MoS₂ devices. There is a significant reduction in the lifetime for each QD ensemble on the MoS₂ devices. The NRET efficiencies, given by $\eta_{NRET} = 1 - (\tau_{QD-MoS_2}/\tau_{QD})$, are $(94 \pm 7)\%$, $(92 \pm 4)\%$, and $(96 \pm 6)\%$, for the 450 nm, 530 nm, and 630 nm QDs, respectively.

The measured photocurrents, I^{Ph} , at a bias of +1 V for each of the devices before and after adding the QD sensitizing layer, as a function of optical excitation power, P_{Ex} , are shown in Fig. 2(d-f). There is a significant increase of the photocurrent of each device after the addition of the QD sensitizing layers. It should be noted that the photocurrent as a function of optical excitation power has the same trend before and after adding the sensitizing layers. Regions corresponding to sub-linear, linear, and super-linear dependence of the photocurrent on the optical excitation power are observed. The sub-linear and linear dependence of photocurrent in MoS₂ devices is commonly observed, however, the super-linear dependence is less reported and is believed to be due to the presence of a variety of recombination centres in the MoS₂ associated with surface imperfections such as surface defects, dangling bonds, and edge defects at grain boundaries [9].

An average MoS₂ photocurrent enhancement, E_{Ph}^{Avg} , is calculated from the low power data, where the photocurrent exhibits a dominant $I^{Ph} = AP_{Ex}^{0.5}$ dependence. This allows for comparison between the photocurrent values and the NRET efficiencies, as the TRPL measurements were performed at low excitation powers (0.2 μ W). The average photocurrent enhancement, $E_{Ph}^{Avg} = (A_{MoS_2-QD}/A_{MoS_2} - 1)$, gives values of 2.9 \pm 0.5, 1.5 \pm 0.1, and 6 \pm 1 for devices with the 450 nm, 530 nm, and 630 nm QD sensitizing layers, respectively. There is good agreement between the trends of NRET efficiency and the average MoS₂ photocurrent enhancement, indicating that the efficiency of the NRET is driving the enhancements in the MoS₂ devices. Similarly, a comparison of the NRET rates calculated from the lifetime data and and the spectral overlap integral reveal almost identical trends, thus confirming that the energy transfer process is that of Förster-type NRET.

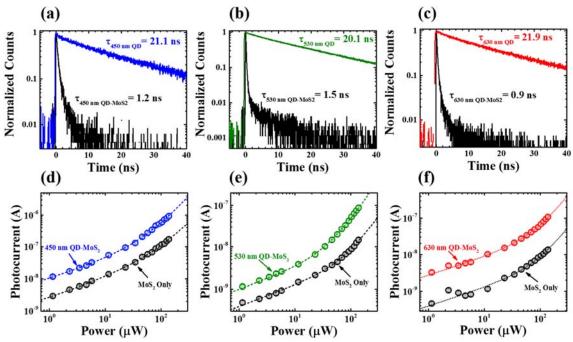


Figure 2: PL decays of (a) 450 nm QDs only (blue) and on the MoS₂ device (black), (b) 530 nm QDs only (green) and on the MoS₂ device, and (c) 630 nm QDs only (red) and on the MoS₂ device. (d-f) Photocurrent measured from the MoS₂ devices before (black) and after adding the QD sensitizing layers.

As the 630 nm QDs show the largest NRET efficiency and lead to the largest photocurrent enhancements in the hybrid devices, these QDs were chosen to study of the dependence of the photocurrent enhancements on the MoS_2 material layer properties. These MoS_2 devices consisted of pristine monolayer (Mono), mixed monolayer/bilayer with a greater proportion of monolayer to bilayer material (Mono/Bi), mixed monolayer/bilayer with a greater proportion of bilayer to monolayer material (Bi/Mono), and polycrystalline bilayer (Poly Bi). It can be seen from the photocurrent plots, presented in Fig. 3(a-d), that the largest enhancements are observed in the monolayer material (\sim 14-fold), with moderate enhancements in the devices consisting of mixed layer material (\sim 2-fold), and no enhancement in the polycrystalline bilayer devices.

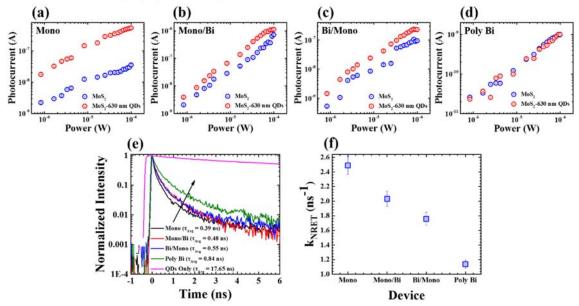


Figure 3: (a-d) Photocurrent as a function of optical excitation power for the MoS_2 devices before (blue) and after (red) adding the QD sensitizing layer. (e) PL decays for the 630 nm QDs on each of the MoS_2 devices. (f) NRET rates for the QDs on the MoS_2 devices.

TRPL measurements were also performed on the QDs on each of these devices, and the PL decays are presented in Fig. 3(e). The shortest QD PL lifetime is measured on the pristine monolayer MoS₂ device and the PL lifetime of the QDs increases as the proportion of MoS₂ in the device channel transitions from pristine monolayer to polycrystalline bilayer, due to the increase in dielectric screening. The NRET rate, k_{NRET} , can be calculated from the QD lifetimes as $k_{NRET} = \tau_{QD-MoS_2}^{-1} - \tau_{QD}^{-1}$. The NRET rates for the four devices, presented in Fig. 3(f), show a more gradual decrease than is observed for the photocurrent enhancement. However, the largest and smallest NRET rates are found for the monolayer and polycrystalline bilayer MoS₂ devices, respectively. This discrepancy between trend of the photocurrent enhancements and the NRET rates is due to the fact that the photocurrent enhancements rely on the transit of charge carriers across the device channel, which can be obstructed by nucleation points and grain boundaries in the MoS₂, while NRET, which is the nonradiative transfer of the QD exciton energy through dipole-dipole coupling, occurs directly between the QDs and the MoS₂ due to the Coulomb interaction between confined excitons in both materials.

4. CONCLUSIONS

In summary, by comparing the NRET rates and the spectral overlap for three hybrid devices consisting of different QD ensembles, we have shown that the mechanism for NRET in hybrid QD-MoS $_2$ device architectures is that of Förster-type NRET. It has also been demonstrated that the photocurrent enhancements in monolayer MoS $_2$ devices are driven by the NRET efficiency from these alloyed QDs. The photocurrent enhancements in the hybrid QD-MoS $_2$ devices are shown to be strongly dependent on the MoS $_2$ material properties. Despite the presence of highly efficient NRET from the QDs for all devices, the photocurrent enhancements are significantly higher for pristine monolayer MoS $_2$. In the case of mixed monolayer/bilayer and polycrystalline MoS $_2$ the presence of nucleation points and defect sites in the MoS $_2$ leads to scattering of charge carriers and subsequently, reduced mobility.

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